

Mechano-Chemical Stability of Gold Nanoparticles Coated with Alkanethiolate SAMs

Brian J. Henz,[†] Takumi Hawa,[‡] and Michael R. Zachariah^{*‡}

U.S. Army Research Laboratory, Aberdeen Proving Ground, Maryland, Departments of Mechanical Engineering and Chemistry and Biochemistry, University of Maryland, College Park, Maryland, and the National Institute of Standards and Technology, Gaithersburg, Maryland

Received August 9, 2007. In Final Form: September 21, 2007

Molecular dynamics simulations are used to probe the structure and stability of alkanethiolate self-assembled monolayers (SAMs) on gold nanoparticles. We observed that the surface of gold nanoparticles becomes highly corrugated by the adsorption of the SAMs. Furthermore, as the temperature is increased, the SAMs dissolve into the gold nanoparticle, creating a liquid mixture at temperatures much lower than the melting temperature of the gold nanoparticle. By analyzing the mechanical and chemical properties of gold nanoparticles at temperatures below the melting point of gold, with different SAM chain lengths and surface coverage properties, we determined that the system is metastable. The model and computational results that provide support for this hypothesis are presented.

Introduction and Background

Nanoparticles are often used in applications where a high surface area to volume ratio is desired. Some of the typical applications include catalysis, biosensors, and use as drug delivery vehicles.^{1,2} A major limitation of the expanded use of these nanoparticles is that fabrication of nanoparticles with the desired morphology and structure is challenging.^{3–5} Gold is a commonly used material because of its resistance to oxidation and interesting electrical, magnetic, optical, and physical properties.⁶ Furthermore, there is a considerable knowledge base on functionalizing gold surfaces that are being ported to analogous particles.⁷ For this reason, gold nanoparticles are commonly chosen for basic research because of the availability of experimental^{8,9} and atomistic simulation data.^{10,11} For instance, using X-ray powder diffraction Cleveland et al.⁹ discovered that small 10–20 Å diameter gold nanoparticles form a truncated-decahedral motif geometry. Using molecular dynamics (MD) simulations Lewis et al.¹² were able to determine the melting process for gold nanoparticles. This last result is important because the authors found that the surface atoms melt first and at a temperature below that of the atoms in the core of the nanoparticle. The implications of this for coated nanoparticles is that the surface atoms, where binding of the monolayer takes place, are affected by a lower temperature than is expected from the melting temperature of

the whole nanoparticle. The results of these past efforts is built upon in this work in order to analyze larger gold nanoparticles, in particular, the nature and role of surface functionalization.

Coating of gold surfaces and nanoparticles with a self-assembled monolayer (SAM) creates many additional applications including lithography, lubrication, catalysis, biocompatible materials, and biosensors.^{6,13} There have been numerous experimental^{14–16} and numerical^{13,17,18} studies performed that have attempted to characterize the properties of alkanethiolate SAM-coated gold surfaces. Simulations investigating alkanethiolate SAM-coated gold surfaces accurately predict the $c(4 \times 2)$ superlattice structure of alkanethiols on an Au(111) surface.¹⁷ Additional work has predicted the segregation of SAM chains adsorbed on a gold surface by length,¹³ surface frictional forces,¹⁹ and phase behavior.²⁰ Efforts to characterize the alkanethiolate SAM-coated gold nanoparticle system have been limited to atomistic simulations considering a single alkanethiol chain and a small (less than 100 atom) gold cluster^{17,21} or MD simulations of small nanoparticles that ignore gold–gold interactions.^{6,18}

During fabrication of nanoparticles the propensity to agglomerate and sinter must be inhibited¹⁸ by passivation of the nanoparticle surface. Use of coatings for surface passivation has been found to significantly reduce the degree to which agglomeration and sintering will occur.²² For instance, Hawa and Zachariah found that a hydrogen surface passivation coating used with a silicon nanoparticle will prevent reaction with other particles when collided at thermal velocities²² or retard the process.²³ Other affects related to the passivation layer investigated

* To whom correspondence should be addressed. Phone: 301-405-4311. Fax: 301-314-9477. E-mail: mrz@umd.edu.

[†] U.S. Army Research Laboratory.

[‡] University of Maryland and NIST.

(1) Manna, A.; Imae, T.; Aoi, K.; Okazaki, M. *Mol. Simul.* **2003**, *29* (10–11), 661–665.

(2) Hong, R.; Han, G.; Fernández, J. M.; Kim, B.-J.; Forbes, N. S.; Rotello, V. M. *J. Am. Chem. Soc.* **2006**, *128*, 1078–1079.

(3) Magnusson, M. H.; Dupert, K.; Malm, J.-O.; Bovin, J.-O.; Samuelson, L. *NanoStruct. Mater.* **1999**, *12*, 45–48.

(4) Chiang, C.-L. *J. Colloid Interface Sci.* **2000**, *230*, 60–66.

(5) Kim, J.-U.; Cha, S.-H.; Shin, K.; Jho, J. Y.; Lee, J.-C. *J. Am. Chem. Soc.* **2005**, *127*, 9962–9963.

(6) Landman, U.; Luedtke, W. D. *Faraday Discuss.* **2004**, *125*, 1–22.

(7) Schreiber, F. *Prog. Surf. Sci.* **2000**, *65*, 151–256.

(8) Greer, J. R.; Nix, W. D. *Appl. Phys. A* **2005**, *80*, 1625–1629.

(9) Cleveland, C. L.; Landman, U.; Schaaff, T. G.; Shafiqullin, M. R.; Stephens, P. W.; Whetten, R. L. *Phys. Rev. Lett.* **1997**, *79* (10), 1873–1876.

(10) Baskes, M. I. *Phys. Rev. B* **1997**, *46* (5), 2727–2742.

(11) Daw, M. S.; Baskes, M. I. *Phys. Rev. B* **1984**, *29* (12), 6443–6453.

(12) Lewis, L. J.; Jensen, P.; Barrat, J.-L. *Phys. Rev. B* **1997**, *56* (4), 2248–2257.

(13) Shevade, A. V.; Zhou, J.; Zin, M. T.; Jiang, S. *Langmuir* **2001**, *17*, 7566–7572.

(14) Nuzzo, R. G.; Zegarski, B. R.; Dubois, L. H. *J. Am. Chem. Soc.* **1987**, *109*, 733–740.

(15) Strong, L.; Whitesides, G. M. *Langmuir* **1988**, *4*, 546–558.

(16) Rosenbaum, A. W.; Freedman, M. A.; Darling, S. B.; Popova, I.; Sibener, S. J. *J. Chem. Phys.* **2004**, *120* (8), 3880–3886.

(17) Zhang, L.; Goddard, W. A., III; Jiang, S. *Comput. Chem. Eng.* **2002**, *22* (10), 1381–1385.

(18) Luedtke, W. D.; Landman, U. *J. Phys. Chem.* **1996**, *100* (32), 13323–13329.

(19) Jiang, S. *Mol. Phys.* **2002**, *100* (14), 2261–2275.

(20) Bhatia, R.; Garrison, B. J. *Langmuir* **1997**, *13*, 765–769.

(21) Masens, C.; Ford, M. J.; Cortie, M. B. *Surf. Sci.* **2005**, *580*, 19–29.

(22) Hawa, T.; Zachariah, M. R. *Phys. Rev. B* **2004**, *69*, 035417.

(23) Hawa, T.; Zachariah, M. R. *Phys. Rev. B* **2005**, *71*, 165434.

computationally include changes in the internal pressure and surface tension.²⁴

This paper is focused on understanding the effect of the alkanethiolate SAM on gold nanoparticles using MD simulations. Using the pair correlation function we estimate the melting temperature of the particle and the effect that the SAM has on the particle surface. We clarify the relationship between surface coverage and structure of the particle by introducing a corrugation factor and identify the phase transition mechanism of SAM-coated gold nanoparticles. We find that the SAM-coated particles are mechanically unstable as determined by the radial pressure profile.

Computational Modeling and Numerical Procedure

There are numerous documented MD studies of alkanethiolate SAMs on gold surfaces.^{13,17,19} However, these studies have been limited to flat gold surfaces or fixed gold atoms in a nanocrystallite.^{18,6} In addition to and in support of these previous MD simulations, ab initio quantum chemical calculations have been used to simulate gold clusters with short alkanethiolate SAMs, primarily to determine potential parameters for MD or MC computations. The ab initio calculations show that the gold crystal lattice is perturbed by the adsorbed ligand.²⁵ The consequence of the perturbation is an increase in Au–Au bond length of up to 20%. The increase in bond length is observed around the adsorption site with relaxation propagating to the second layer. These results suggest that the MD simulations should include dynamic gold atoms in order to accurately consider the nanoparticle/SAM system.

Alkanethiol chains adsorbed on a flat surface made of immobile gold atoms are limited to diffusion in only two dimensions, whereas chains adsorbed onto a nanoparticle surface with mobile gold atoms may diffuse in three dimensions. This disparity in model detail is expected to provide more insight into the SAM-coated nanoparticle system than previously available. Inclusion in the simulation of a surface constructed of dynamic gold atoms should be able to more accurately predict alkanethiol chain movement than the limited movement possible on a flat surface of fixed gold atoms.

The size of the model system investigated here is determined by approximating experimentally realistic systems. The manufacture of consistently sized small diameter nanoparticles becomes more difficult at diameters of less than 50 Å; therefore, in this work 50 Å diameter gold nanoparticles (4093 gold atoms) with and without SAMs are considered. The SAM considered consists of alkanethiolate chains that contain a sulfur atom head group that binds to the gold surface and a carbon backbone with 3–18 carbon atoms comprising the alkyl chain.

The MD simulations in this work are carried out using the LAMMPS (Large-scale Atomic/Molecular Massively Parallel Simulator)²⁶ software.

Simulating the Gold Substrate. In this effort the gold substrate atoms are explicitly considered as dynamic atoms, requiring that an accurate potential be used to simulate these atomic interactions. The embedded atom method^{11,27,28} (EAM) and the effective medium^{29–31} (EM) potentials have received the most attention for accurately simulating atomic gold interactions and the interactions of other fcc lattice metals. Both of these methods use empirical embedding functions that originate from density functional theory (DFT) and have been successfully used to model many of the physical

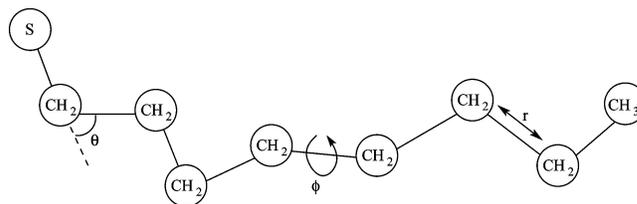


Figure 1. Sample alkanethiolate chain with 8 carbon atoms along the backbone.

properties of gold. We chose the EAM for modeling the gold substrate because of its reported accuracy and the potentially important extensions developed for the EAM for modeling surfaces, such as the extended EAM³² (XEAM) and the surface EAM (SEAM).³³ The actual EAM potential used in this work is taken from LAMMPS.²⁶ Also considered was a second dataset from Voter and Chen.³⁴ A detailed discussion of the method as used in this work can be found in Daw et al.^{11,27}

Modeling the Alkanethiolate Chain. The alkanethiolate polymer chain may be simulated with a range of computational complexity depending upon the resources available and the accuracy of results required. For MD simulations, the three methods available for modeling the alkanethiol chain are the high-accuracy/high-computational requirement all-atom model, the less accurate/lower computational requirement united atom method, and the coarse-grained bond model. These methods can be compared by considering the CH₂ molecule that makes up the backbone of the alkanethiolate chain. When using the all-atom model, the numerical simulation must consider three particles for each molecule and the interactions between each. Using the united atom model only one particle is simulated for each molecule,³⁵ which eliminates the intermolecular interactions and limits the number of intermolecular interaction potentials required. Finally, with the coarse-grained model, multiple molecules are clustered together into one particle with perhaps five or more CH₂ molecules grouped into one simulated particle.^{36–38} For the simulations in this work we have chosen the united atom method, which is computationally more efficient than the all-atom method yet provides sufficient accuracy for the current analysis.¹⁸

The modeling of the alkanethiol chain using the united atom model is consistent with many of the published studies for the alkanethiolate-coated gold nanoparticle system.^{6,17,18,20} An example of the alkanethiolate polymer chain model used in this work is shown in Figure 1. The potentials used in the united atom model include a bending potential for the S–C–C and C–C–C bonds. Bond bending can be modeled as either a flexible bond or a stiff bond where the angle is held constant. There is a dihedral potential for the X–C–C–X bonds, where X can be either S or C. There are two common methods used in the literature to model the bond lengths along the chain. The bond lengths along the chain can be held constant¹³ with the RATTLE³⁹ algorithm or they can be modeled as harmonic bonds.^{40,41} Using a harmonic bond to describe the C–C and S–C backbone bonds allows the distance between the backbone atoms to change while not allowing the bonds to break during the simulation.

(24) Hawa, T.; Zachariah, M. R. *J. Chem. Phys.* **2004**, *121* (18), 9043–9049.

(25) Grönbeck, H.; Curioni, A.; Andreoni, W. *J. Am. Chem. Soc.* **2000**, *122*, 3839–3842.

(26) Plimpton, S. J. *J. Comput. Phys.* **1995**, *117*, 1–19.

(27) Foiles, S. M.; Baskes, M. I.; Daw, M. S. *Phys. Rev. B* **1986**, *33* (12), 7983–7991.

(28) Mei, J.; Davenport, J. W.; Fernando, G. W. *Phys. Rev. B* **1991**, *43* (6), 4653–4658.

(29) Stave, M. S.; Sanders, D. E.; Raeker, T. J.; DePristo, A. E. *J. Chem. Phys.* **1990**, *93* (6), 4413–4426.

(30) Raeker, T. J.; DePristo, A. E. *Int. Rev. Phys. Chem.* **1991**, *10* (1), 1–54.

(31) Kelchner, C. L.; Halstead, D. M.; Perkins, L. S.; Wallace, N. M.; DePristo, A. E. *Surf. Sci.* **1994**, *310*, 425–435.

(32) Lee, B.; Cho, K. *Surf. Sci.* **2006**, *600*, 1982–1990.

(33) Haftel, M. I. *Phys. Rev. B* **1993**, *48* (4), 2611–2622.

(34) Voter, A. F.; Chen, S. P. In *Characterization of Defects in Materials*; Materials Research Society: Pittsburgh, 1987; Vol. 82, p 175.

(35) Jorgensen, W. L.; Madura, J. D.; Swenson, C. J. *J. Am. Chem. Soc.* **1984**, *106*, 6638–6646.

(36) Fukunaga, H.; Takimoto, J. I.; Doi, M. *J. Chem. Phys.* **2002**, *116* (18), 8183–8190.

(37) Tries, V.; Paul, W.; Baschnagel, J.; Binder, K. *J. Chem. Phys.* **1997**, *106* (2), 738–748.

(38) Maiti, P. K.; Lansac, Y.; Glaser, M. A.; Clark, N. A. *Langmuir* **2002**, *18*, 1908–1918.

(39) Andersen, H. C. *J. Comput. Phys.* **1983**, *52*, 24–34.

(40) Rai, B.; Sathish, P.; Malhotra, C. P.; Pradip Ayappa, K. G. *Langmuir* **2004**, *20*, 3138–3144.

(41) Mahaffy, R.; Bhatia, R.; Garrison, B. J. *J. Phys. Chem. B* **1997**, *101*, 771–773.

The potentials and parameter values for the bending angle and dihedral angle from Figure 1 and used in this work are given in Shevade et al.;¹³ the bond stretch parameters are given in Rai et al.⁴⁰

Binding of Sulfur to Gold. The binding potential of the head sulfur atom of an alkanethiol chain to the gold substrate is of great interest as it affects the location, orientation, movement, and desorption of chains from the gold surface.^{6,41,42} There has been much effort devoted to finding accurate potentials for simulating the sulfur to gold binding energy from ab initio methods.^{6,42} The most commonly published potentials resulting from these calculations are the 12-3 potential,⁴³ used most commonly in Monte Carlo (MC) simulations of alkanethiolates on gold^{13,19} and the Morse potential used in many of the MD simulation studies.^{19,17,41,44} In addition to the issue of which potential best describes the binding between SAM headgroup and substrate, the possibility of SAM mobility must also be considered.⁴² Since the Morse potential, eq 1, can mimic a partially covalent bond,¹⁷ it is used in MD simulations along with the parameters given by Zhang et al.¹⁷ to model the Au–S bond. The parameters for eq 1 are $D_e = 13.3$ kJ/mol, $\alpha = 1.38$, and $r_0 = 2.903$ Å.

$$U_{\text{Morse}}(r) = D_e [e^{-2\alpha(r-r_0)} - 2e^{-\alpha(r-r_0)}] \quad (1)$$

The binding energy of the alkanethiol to the gold surface has been determined to be around 184.1 kJ/mol¹³ when all of the sulfur–gold interactions are considered. Although this reported value is often referenced, there are several studies that report the sulfur–gold bond energy to be closer to 126.0 kJ/mol.^{45,46} This discrepancy in the binding energy is large, and so in this work we will investigate using both binding energies in order to understand the affect that binding energy has on the computed results. For the sulfur–gold binding energy of 126.0 kJ/mol, the Morse potential parameter $D_e = 9.108$ kJ/mol.

Simulation Results

In this section, the results from the MD simulations are presented and analyzed. In order to study the affects of the alkanethiolate SAM on the gold nanoparticle we computed many system properties. These properties include diffusion coefficients, radial pressure and density distributions, melting temperature, and the pair correlation function. A corrugation factor that describes the depth to which the gold surface is modified by the adsorbed alkanethiol chains is also defined in this section. Each of these results is a data point of the changes the nanoparticle surface experiences with the adsorption of the alkanethiolate SAM. For instance, the diffusion coefficient is used to compute the mobility of the SAM chains that are adsorbed on the surface of the nanoparticle and the mobility of the gold atoms near the surface of the nanoparticle. Combination of these two coefficients will partially define the phase of the materials at and on the nanoparticle surface and also help in predicting if nucleation of the SAM chains will occur. Nucleation may occur under low surface coverage conditions when the chains form dense groups and uneven surface coverage. The density distribution is used to find the depth to which the sulfur atoms penetrate the surface of the gold nanoparticle and compute the corrugation factor.

The simulation results will also be compared for the two sulfur–gold binding energies considered in this work. The computed results will be compared and discussed in order to better

understand how the head group binding energy affects the properties of the nanoparticle and anticipated trends for these properties. It is expected that the lower binding energy will have less of an affect on the properties of the gold nanoparticle than the high binding energy. On the other hand, some processes such as desorbtion will possibly be observed only at the lower binding energy because the temperature required for desorbtion will be lower.

The simulation is initialized with a sphere of fcc lattice gold atoms at 0 K. By applying a random velocity distribution to the gold atoms the temperature is raised to 10 K and then slowly increased over 10^7 time steps to 1400 K or above the bulk melting temperature of gold. By rescaling the velocity the temperature is then decreased to 300 K over 8×10^6 time steps and then held for another 8×10^6 time steps. The temperature rescaling fix is then removed in order to determine if the total energy of the system remains constant. The nanoparticle now contains many large facets with (111) and (100) faces, as one would expect from experimental observations.⁴⁷ The alkanethiol chains are next randomly distributed around the nanoparticle with the sulfur head group at the equilibrium distance from the gold surface. The alkanethiol chains are initially orientated radially from the nanoparticle center. The alkanethiol chains were initialized with 100% gauche defects. The affect of this initialization on the observed simulation results is negligible as many of the gauche defects are removed during equilibration. The only result that would not be considered accurate is a quantitative measurement of the fraction of gauche defects under various conditions. The number of chains is determined by the requested surface area per chain. The temperature is again initialized to 10 K, slowly heated to 300 K, and then held until the total system energy remains constant, indicating that equilibrium has been achieved. The temperature during equilibration of the alkanethiol-coated nanoparticle is not allowed to increase beyond 300 K because desorbtion is expected to occur beyond this temperature.⁷

Pair Correlation Function

The pair correlation function, $g(r)$, is defined as the number of atoms a distance r from a given atom compared with the number of atoms at the same distance in an ideal gas at the same density.⁴⁸ The pair correlation function for a bare 50 Å diameter gold nanoparticle is shown in Figure 2 for a range of temperatures. The results in Figure 2a indicate that a phase transition occurs in the nanoparticle from a crystalline solid at 900 K to a liquid at 1000 K, and as expected, this temperature range is below that of the bulk melting temperature of 1340 K.⁴⁹

For a spherical nanoparticle the melting temperature can be crudely predicted from experimental data using eq 2, where ρ_S and ρ_L are the specific mass of the solid and liquid phases, respectively. γ_S and γ_L are the surface energies of the solid and liquid phases, respectively, D_p is the particle diameter, L is the heat of fusion, and T_∞ and $T(D_p)$ are the bulk and size-dependent melting temperatures, respectively. Using published data for gold⁴⁹ the predicted melting temperature of a 50 Å diameter gold nanoparticle is about 995 K. This result is within

(42) Beardmore, K. M.; Kress, J. D.; Gronbech-Jensen, N.; Bishop, A. R. *Chem. Phys. Lett.* **1998**, *286*, 40–45.

(43) Hautman, J.; Klein, M. L. *J. Chem. Phys.* **1989**, *91* (8), 4944–5001.

(44) Liu, K. S. S.; Yong, C. W.; Garrison, B. J.; Vickerman, J. C. *J. Phys. Chem. B* **1999**, *103*, 3195–3205.

(45) Lavrich, D. J.; Wetterer, S. M.; Bernasek, S. L.; Scoles, G. *J. Phys. Chem. B* **1998**, *102*, 3456–3465.

(46) Jang, S. S.; Jang, Y. H.; Kim, Y.-H.; Goddard, W. A., III; Flood, A. H.; Laursen, B. W.; Tseng, H.-R.; Stoddart, J. F.; Jeppesen, J. O.; Choi, J. W.; Steurman, D. W.; Delonno, E.; Heath, J. R. *J. Am. Chem. Soc.* **2005**, *127*, 1563–1575.

(47) Schmid, G. *Chem. Rev.* **1992**, *92*, 1709–1727.

(48) Allen, M. P.; Tildesley, D. J. *Computer Simulation of Liquids*; Oxford University Press Inc.: New York, 1987.

(49) Lewis, L. J.; Jensen, P.; Barrat, J.-L. *Phys. Rev. B* **1997**, *56* (4), 2248–2257.

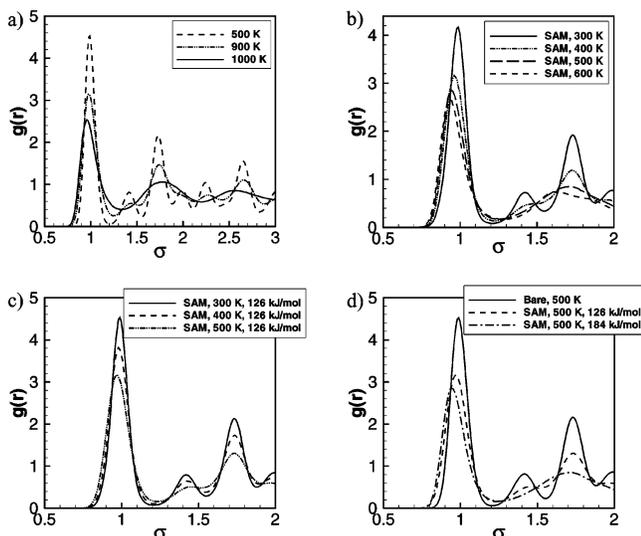


Figure 2. (a) Pair correlation function for a bare 50 Å gold nanoparticle. Note the change in shape and location of peaks between 900 and 1000 K indicating a phase change from solid to liquid. (b) Pair correlation function results for the gold atoms in a 50 Å gold nanoparticle coated with an alkanethiolate SAM at various temperatures with a sulfur binding energy of 184 kJ/mol. (c) Pair correlation function results for the SAM-coated gold nanoparticle with the lower sulfur binding energy of 126 kJ/mol. (d) Comparison of the pair correlation results at 500 K for the bare nanoparticle and high (184 kJ/mol) and low (126 kJ/mol) alkanethiol binding energies.

the range of our MD simulation prediction for the melting temperature.

$$\frac{T(D_p) - T_\infty}{T_\infty} = \frac{4}{\rho_s L D_p} [\gamma_s - \gamma_L (\rho_s / \rho_L)^{2/3}] \quad (2)$$

Next, we compare the pair correlation function results for the uncoated gold nanoparticle to the results for an alkanethiolate-coated gold nanoparticle. The coating considered is a full coating, meaning one chain per 15.4 Å² of surface area on the gold nanoparticle,⁵⁰ and is about 30% denser than the packing density on a flat surface (21.4 Å²).¹⁵ The pair correlation function results for the alkanethiolate SAM-coated gold nanoparticle show a nanoparticle with a crystalline structure for temperatures below 400 K and an amorphous structure above 600 K, see Figure 2b. The pair correlation function is computed for the gold atoms only in Figure 2b. By only considering the gold atoms in computing the pair correlation results, the structure of the nanoparticle is analyzed separately from the SAM. At 500 K, the structure of the gold nanoparticle has changed to a less ordered structure, similar to that observed during the melting of the uncoated gold nanoparticle. This change in the pair correlation function indicates that a phase change occurs in the gold nanoparticle between 400 and 500 K. Hence, the melting temperature of the gold nanoparticle is lowered by about 500 K by mixing with the alkanethiolate chains.

In the next set of simulations we considered the lower gold–sulfur binding energy of 126 kJ/mol. In these simulations the chain–chain interactions use the same interaction potentials discussed earlier. In Figure 2c the pair correlation results for the fully coated gold nanoparticle are plotted for temperatures of 300, 400, and 500 K. Notice that at 500 K the nanoparticle is still highly structured. This indicates that the nanoparticle does not undergo a phase change at 500 K as is observed in the high

binding energy results. In Figure 2d the pair correlation results for the gold nanoparticle at 500 K are compared for the uncoated and coated nanoparticles with high and low binding energy. In this figure the affect of the high binding energy is clearly evident. The lower degree of crystallinity with the higher binding energy indicates a more liquid-like structure at 500 K.

Diffusion

From the results of the previous section it is apparent that the alkanethiolate SAM coating has an appreciable affect on the properties of the gold nanoparticle. In order to gain a better understanding of the SAM and how it interacts with the gold nanoparticle we computed the diffusion coefficient for the SAM and the gold atoms. The alkanethiolate chain is primarily bound to the gold nanoparticle through the sulfur–gold interactions at the head of the chain. The diffusion coefficient of the sulfur atoms is of particular interest because it will provide insight into the mobility of the alkanethiol chains in relation to the surface gold atoms. The diffusion coefficient of the alkanethiolate chain is calculated in the MD simulations by computing the mean squared displacement of the sulfur head group. The diffusion coefficient, D , is then evaluated using eq 3.

$$\frac{\partial \langle r^2(t) \rangle}{\partial t} = 2dD \quad (3)$$

In eq 3, the number of dimensions available for atomic diffusion, d , is assumed to be 3, t is time, and $\langle r^2(t) \rangle$ is the mean squared displacement (MSD) of the atoms being tracked. For diffusion on a surface, d would typically be 2. However, for a nanoparticle, as opposed to a flat surface, the gold atoms near the nanoparticle surface are less strongly bound to the core atoms and thus more mobile, allowing for the sulfur atoms to move in a third dimension normal to the nanoparticle surface. This movement normal to the nanoparticle surface has not previously been considered in the simulation of SAM-coated gold nanoparticles but is expected for the reasons discussed previously.

The diffusivity of the alkanethiol chain is computed by considering the movement of the sulfur head group. By only using the displacement of the sulfur head group our results are not affected by extraneous data, such as the movement of the alkanethiol backbone. In addition to studying the diffusivity of the sulfur atoms, the diffusivity of the surface gold atoms that are bound to the sulfur atoms is also calculated. By comparing the diffusivity of the surface gold atoms with that of the sulfur atoms it is possible to determine if the alkanethiolate chains are sliding over or moving with the gold surface atoms. For example, if the diffusivity of the sulfur atoms is very different from the gold atoms then the sulfur atoms are moving over the nanoparticle surface, the SAM chains are considered mobile, and they may nucleate. Using eq 3, the diffusion coefficients for the sulfur head group and surface gold atoms are computed to be 8.05×10^{-7} and $9 \times 75 \cdot 10^{-8}$ cm²/s, respectively. These diffusion coefficients are computed for a fully coated 50 Å gold nanoparticle at 300 K. The difference in diffusion coefficients is about 1 order of magnitude, indicating that the alkanethiolate SAM is relatively mobile on the nanoparticle surface. From observations of longer simulations (> 10 ns), with low surface coverage, there is no indication of nucleation of the SAM chains. This may be expected because of the attractive potential between the alkanethiolate chain and the gold surface, so that the chain monomers prefer to bond with as many gold atoms as possible in order to minimize the system energy. As such, we see no evidence at low surface coverage of free-standing groups of chains.

(50) Badia, A.; Cuccia, L.; Demers, L.; Morin, F.; Lennox, R. B. *J. Am. Chem. Soc.* **1997**, *119*, 2682–2692.

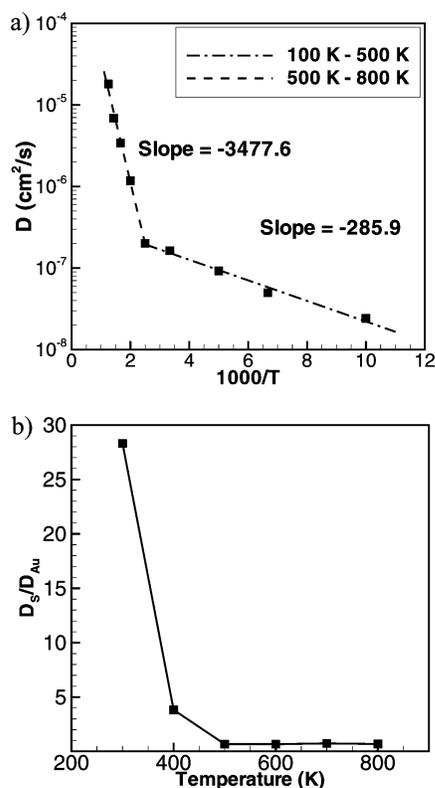


Figure 3. (a) Arrhenius plot of diffusivity for sulfur atoms. The activation energy for SAM mobility is estimated by the slope of $\ln(D)$ vs $1/T$ between 100 and 800 K. (b) Ratio of sulfur diffusivity to gold diffusivity in the alkanethiol-coated gold nanoparticle material system.

The diffusion coefficient of the sulfur head group has been computed for various system temperatures and is presented in an Arrhenius plot in Figure 3. There are two linear regions with slopes of -285.9 and -3477.6 , indicating two distinct phases of the alkanethiolate SAM in the temperature ranges of 100–500 and 500–800 K, respectively.

The activation energy for the low- and high-temperature regions is 2.4 and 29.0 kJ/mol, respectively. This is the expected behavior for heterodiffusion, where one type of atom diffuses on another.^{51,52} In the low-temperature regime the observed activation energy corresponds to the alkanethiol chains hopping between adsorption sites, which is observed as diffusion. For flat gold surfaces, particularly the Au(111) surface, the adsorption binding energy surface has been computed using atomistic simulations.^{17,53} These atomistic computations have determined that the difference between the minimum binding energy (face-centered cubic) site and the maximum binding energy (atop) site is between 25.1 and 16.9 kJ/mol. The diffusion barrier is the energy required for an adsorbate atom to move to the next adsorption site. In ref 17 Zhang et al. calculated the binding energy at a bridge site that would provide a possible diffusion path between adsorption sites with a diffusion barrier of 12.2 kJ/mol. Each of these published results is for flat surfaces and reports a somewhat higher activation energy than the computed activation energy from the MD simulations for the low-temperature regime. One possible explanation for the discrepancy in the calculated value is that for a curved nanoparticle surface there may be lower energy and irregular diffusion paths available for the alkanethiolate chains

to diffuse along. More likely however is that all the observed activation energies are small, and in part, at least differences may be associated with the accuracy of the interatomic potential. Nevertheless, it does indicate a significant difference between a low-temperature diffusion process and that occurring at higher temperature.

Typically, as the temperature rises the dominant mechanism for diffusion shifts from adatoms to vacancies. However, what is occurring here is that the diffusion of the alkanethiolate chain is dominated by dissolution into the gold nanoparticle, resulting in a more highly activated process. In both instances the activation energy required for diffusion is increased, but in the former process the change in activation energy is observed to be a factor of 2 or more.⁵¹ In this case the change is about 1 order of magnitude. The driving force for dissolution will be discussed in the next section. We notice however, in Figure 3b, that the diffusivity of the gold atoms is more than an order of magnitude lower than the sulfur head group atoms at low temperatures, below 500 K. This difference suggests that the alkanethiol chains are moving freely over the gold nanoparticle surface, but at around 500 K this is no longer the case, and the diffusivity of the sulfur atoms closely tracks that of the gold nanoparticle atoms. Above 500 K the diffusivity of the gold and sulfur atoms is similar, supporting the observation that mixing of the two materials has occurred.

Radial Pressure Distribution

The radial pressure distribution is used in this work to quantify the affect of the adsorbed alkanethiolate SAM on the gold nanoparticle. In very small droplets the internal pressure can be much larger than the surrounding environment. One method of evaluating the radial pressure distribution is to use the normal component of the Irving–Kirkwood (IK) pressure tensor.⁵⁴ The IK pressure tensor comprises two terms, corresponding to a kinetic $P_K(r,T)$ and a configurational $P_U(r,u)$ contribution as shown in eq 4.

$$P_N(r) = P_K(r,T) + P_U(r,u) \quad (4)$$

The kinetic pressure term is a function of temperature and the computed radial density distribution. The configurational term is computed from the interactions, u , between pairs of particles. For each shell of radius, r , the forces between particles whose line of interaction intersects the shell is considered when computing the normal pressure component. The normal pressure at the surface of a nanoparticle or droplet is typically positive, indicating a compressive surface tension. The radial pressure distribution for an uncoated 50 Å gold nanoparticle as a function of temperature is plotted in Figure 4.

With the gold nanoparticle melting temperature between 900 and 1000 K, as discussed in the pair correlation function results, the “Bare, 1000 K” curve is for liquid gold and the other curves are for a solid gold nanoparticle. The radial pressure profile below 900 K shows oscillations since the particle is in the solid phase. On the other hand, the oscillation of the pressure profile at 1000 K disappears because of greater atom mobility and indicates a phase transition in agreement with the pair correlation results previously discussed. As the temperature of the nanoparticle increases, the surface pressure decreases. This decrease of pressure with increasing temperature is expected because the reported surface tension data decreases with increasing temperature. Also, from an analysis of the terms in the IK pressure tensor, eq 4, it is apparent that if the density remains relatively

(51) Dalton, A. S.; Seebauer, E. G. *Surf. Sci.* **2007**, *601*, 728–734.

(52) Suni, I. I. *Surf. Sci.* **1995**, *309*, L179–L183.

(53) Sellers, H.; Ulman, A.; Shnidman, Y.; Eilers, J. E. *J. Am. Chem. Soc.* **1993**, *115*, 530–542.

(54) Thompson, S. M.; Gubbins, K. E.; Walton, J. P. R. B.; Chantry, R. A. R.; Rowlinson, J. S. *J. Chem. Phys.* **1984**, *81* (1), 530–542.

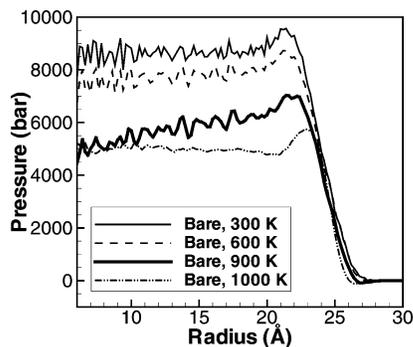


Figure 4. Radial pressure distribution for 50 Å gold nanoparticle at various temperatures.

constant then the kinetic pressure term will increase with increasing temperature. This means that the configurational term is decreasing at an even greater rate. The magnitude of the configurational term decreases when the interactions between the gold atoms decreases or are attractive. The fact that the configurational pressure term is negative indicates that the interactions between the atoms are primarily attractive. The attractive potential is due to the fact that the embedding term in the EAM potential is greater in magnitude than the core repulsive term. This occurs when the gold–gold bonds are stretched as occurs during heating. Additionally, if the average coordination number of the gold atoms were to increase with increasing temperature, as observed in silicon,⁵⁵ then the electron density would increase. This increase in electron density causes the attractive embedding term to increase while the repulsive term, which is evaluated on a pairwise basis, would not change.

From a dimensional analysis of the Young–Laplace⁵⁶ theorem shown as eq 5, this expected result is confirmed.

$$\Delta P = \frac{2\gamma}{r} \quad (5)$$

In eq 5, r is the radius of the gold nanoparticle and ΔP is the difference in pressure between the interior and exterior of the droplet, i.e. $P_{\text{inside}} - P_{\text{outside}}$. Experimentally determined and published values for gold surface tension are wide ranging and available from a limited number of sources. For the solid–gas interface the surface tension, γ_{sg} , is given as 1400 $\text{mJ}\cdot\text{m}^{-2}$, and for the liquid–gas interface the surface tension, γ_{lg} , is 1128 $\text{mJ}\cdot\text{m}^{-2}$.⁵⁶ For a vacuum interface, γ_{lv} is about 742 $\text{mJ}\cdot\text{m}^{-2}$ at the melting point of gold.⁵⁷ It could be expected that the difference in measured surface tension between the liquid–gas and liquid–vacuum would be minimal. From the two published sources this is not the case, and so for comparison purposes we must consider these values as representative of the expected value. Evaluation of eq 5 for our 25 Å radius particle and the published surface tension data for a vacuum interface results in a predicted internal pressure of ~6000 bar. This analytic result is close to that obtained from the MD simulation at 1000 K shown in Figure 4. With the wide range of available data for comparison it can only be said that the predicted surface pressure and simulated pressure are in reasonable agreement.

One can, of course, use the pressure profile from the MD simulation to directly compute the surface tension using the

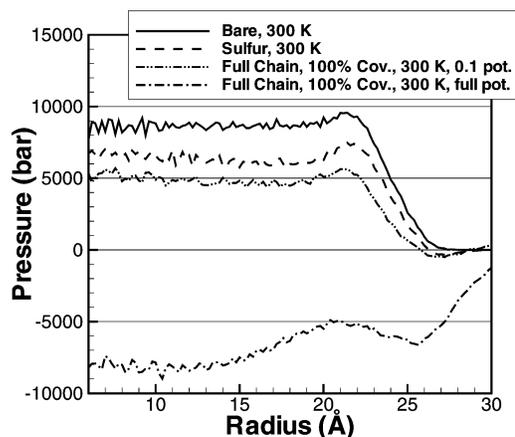


Figure 5. Radial pressure distribution for 50 Å gold nanoparticle with alkanethiolate SAMs with differing interaction coefficients.

approach outlined by Rowlinson and Widom⁵⁸ and presented in eq 6.

$$\gamma_i^3 = -\frac{1}{8}(p_{r_2} - p_{r_1})^2 \int_{r_1}^{r_2} r^3 \frac{dp_N(r)}{dr} dr \quad (6)$$

where p_{r_2} and p_{r_1} are the computed pressure at the maximum and minimum model radii, respectively, and $dp_N(r)/dr$ is the gradient of the normal pressure at radius r . Using this model, the surface tension values calculated for the bare 50 Å gold nanoparticle at 900 and 1000 K are 751 and 617 mN/m , respectively. These temperatures bound the melting temperature of the gold nanoparticle, and the surface tension results bound the published data of 742 mN/m .

Combination of the results from the Young–Laplace and Rowlinson–Widom equations shows that the internal pressure computations for the uncoated gold nanoparticle are reasonable and in the range expected. Unfortunately, there is no data available for comparison when considering the alkanethiolate-coated gold nanoparticle surface.

From past efforts²³ it is expected that a surface coating will modify the nanoparticle internal pressure, through the surface tension, of the gold nanoparticle. This possibility is considered here by comparing the radial pressure distribution for gold nanoparticles with varying degrees of surface coatings. The systems considered include a bare gold nanoparticle, a gold nanoparticle with only sulfur atoms adsorbed onto the surface, and finally a gold nanoparticle with alkanethiolate chains adsorbed on the surface but with varying interaction potentials. For bare and coated gold nanoparticles with weak SAM chain–chain interactions the internal pressure is positive, meaning a positive surface tension. This result is expected, and the trend is also reasonably expected from past research involving hydrogen-passivated silicon nanoparticles.²³ As observed in Figure 5, the pressure trends lower and becomes negative on the interior of the fully coated nanoparticle as the chain–chain and chain–gold interaction potential parameter values are increased to the published values, indicated by “Full Chain, 100% Cov., 300 K, full pot.”.

As a first interpretation of this negative pressure result, one could consider the volume integral of the internal pressure distribution. Since the volume integral of the pressure distribution is negative, it indicates a mechanically unstable system or a nanoparticle under tension. As seen in Figure 5, the negative

(55) Zachariah, M. R.; Carrier, M. J.; Blaisten-Barojas, E. *J. Phys. Chem.* 1996, 100, 14856–14864.

(56) Winn, J. S. *Physical Chemistry*; Addison Wesley: Boston, MA, 1995.

(57) Lang, G. *Handbook of Chemistry and Physics*, 55th ed.; CRC Press: Cleveland, OH, 1974.

(58) Rowlinson, J. S.; Widom, B. *Molecular Theory of Capillarity*; Dover Publications Inc.: Mineola, NY, 1982.

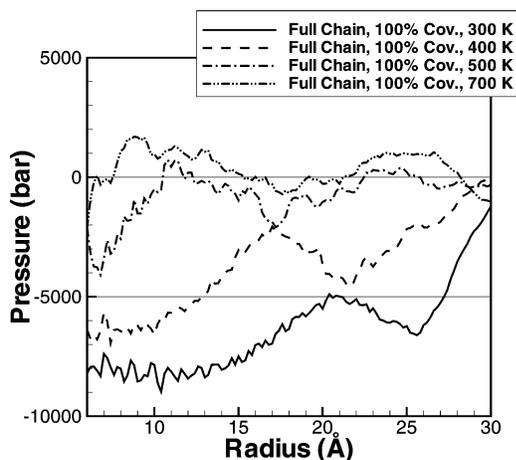


Figure 6. Radial pressure distribution for fully coated gold nanoparticle for a range of temperatures with high binding energy (184.1 kJ/mol).

pressure is observed in the interior of the nanoparticle and only when full chain–chain and chain–gold interactions are considered. This negative pressure result indicates that the gold nanoparticle is experiencing a tensile stress and will change its configuration over time or with increases in temperature in order to accommodate the positive stress or negative pressure. The chain–chain interactions apparently have a strong influence on the surface tension and, therefore, on the internal pressure of the nanoparticle. The strength of the interaction potential that will flip the internal pressure from positive and mechanically stable to negative and mechanically unstable is still under investigation but is between 10% and 100% of the published potentials used for the alkanethiolate SAM interaction potential. Other factors that may affect the surface tension are the surface area per adsorbed alkanethiol chain and temperature. In Figure 6 the radial pressure profile is plotted for a fully coated gold nanoparticle with a sulfur–gold binding energy of 184.1 kJ/mol. Notice that as the temperature increases the pressure that is primarily negative at 300 K becomes positive at 700 K. As the system is held at a temperature above the melting temperature of the solution the pressure continues to increase and will most likely equilibrate to an average pressure above 0.

A negative surface tension resulting in a tensile stress inside the nanoparticle is not stable for a liquid nanoparticle and will result in deformation. In a solid nanoparticle the tensile stress will exist for a time before creep⁵⁹ or an increase in temperature will lower the yield strength allowing the nanoparticle to yield and flow. For these reasons the solid gold nanoparticle with an alkanethiolate coating is metastable at low temperatures but will stabilize at higher temperatures by changing shape. In this particular case we observed, as discussed in the section on diffusion, that the alkylthiol chains dissolve into the particle. This result has not been previously observed in computer simulations because the gold nanoparticle was assumed to maintain its shape with the interaction of the alkanethiolate SAM¹⁸ constrained to the surface. Using the lower, 126 kJ/mol, binding energy we initially observe desorption of alkanethiolate chains from the nanoparticle surface as the temperature is increased until the temperature is above 600 K, at which time the remaining alkanethiolate chains dissolve into the nanoparticle.

In Figure 7 the internal pressure of the alkanethiol-coated gold nanoparticle is plotted from simulation data collected with a sulfur–gold binding energy of 126 kJ/mol.

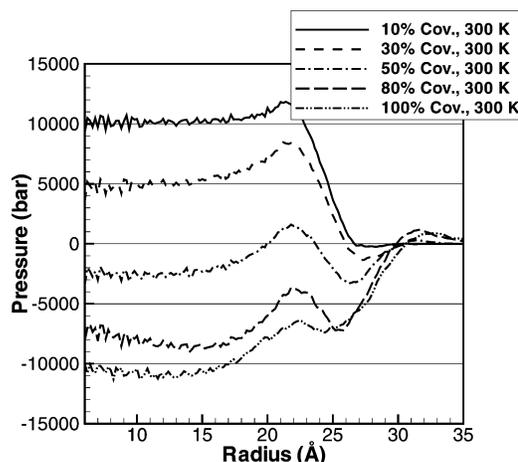


Figure 7. Radial pressure distribution for an alkanethiol-coated 50 Å gold nanoparticle with a 126 kJ/mol sulfur–gold binding energy.

Notice in Figure 7 that the internal pressure of the fully coated nanoparticle with the 126 kJ/mol sulfur binding energy is very similar to the internal pressure of the fully coated gold nanoparticle in Figure 5, where the binding energy of the sulfur–gold bond is 184 kJ/mol. Recall that only the sulfur–gold binding energy has been changed in Figure 7, whereas in Figure 5 the “0.1 pot.” curve represents data collected from simulations where the sulfur–gold and the chain–chain interactions have both been modified. By comparing these results we can conclude that the chain–chain interactions have a greater affect on internal pressure than the sulfur–gold binding energy. This is an important result because although the sulfur–gold binding energy has some affect on the structure of the gold nanoparticle and diffusivity, it is not the primary factor in determining the stability of the nanoparticle. Additionally, the affect of chain length on the radial pressure distribution is also considered. After computing the radial pressure distribution for alkanethiol chain lengths of 3 and 18 carbon atoms, we found that the shorter chains result in an internal pressure about 5000 bar higher than the previously considered 9 carbon atom chains. Alternatively, the longer chains do not appear to have a greater affect on the internal pressure of the nanoparticle than the 9 carbon atom chains studied here. This leads to the conclusion that for the number of monomers considered a longer chain has little affect on the stability of the nanoparticle, whereas a shorter chain will lower the total chain–chain interaction sufficiently to affect the nanoparticle stability.

Radial Density Distribution

From the computed results up to this point we determined that the 50 Å uncoated gold nanoparticle has a melting temperature between 900 and 1000 K (pair correlation function), the SAM chains are mobile on the gold surface (diffusion coefficient), and there are two phases for the adsorbed alkanethiolate SAM between 100 and 800 K. In this section the radial density results are analyzed. The radial density is computed as part of the radial pressure calculation, discussed previously. For bare nanoparticles, the radial density distribution for the gold atoms will change abruptly at the nanoparticle surface from the core density to zero. On the other hand, for coated nanoparticles, if the surface atoms move radially outward due to the presence of the alkanethiolate chains penetrating into the particle, the slope of the radial density distribution will be less steep. This phenomenon of the SAM chains penetrating into the nanoparticle surface is not a previously predicted result because past efforts to simulate alkanethiolate-coated gold nanoparticles assumed that the gold atoms do not move as a result of the interactions with the SAM.^{18,40}

(59) Dieter, G. E. *Mechanical Metallurgy*; McGraw-Hill Inc.: New York, 1986.

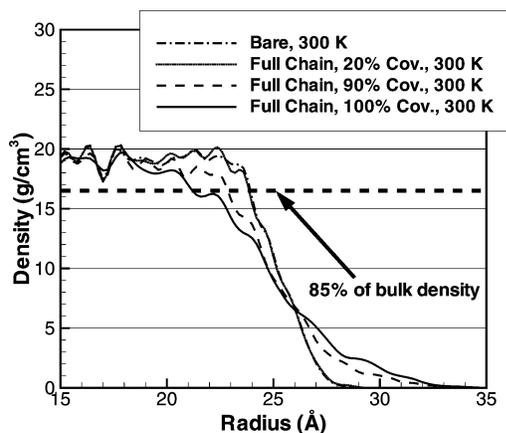


Figure 8. Radial density of gold in an alkanethiolate SAM-coated gold nanoparticle for various surface coverage amounts.

These previous results have either assumed a flat surface or a nanocrystallite with immobile atoms. Modeling the gold atoms using the EAM potential provides for an investigation of whether or not the gold atoms are affected by the SAM. When analyzing the radial density, shown in Figure 8, it is apparent that the atoms near the surface of the gold nanoparticle are less densely packed than those in the nanoparticle core. This decrease in gold atomic density suggests that near the nanoparticle surface the gold atoms are mixing with the alkanethiolate molecules.

The results in Figure 8 show the expected steep slope for the radial density distribution of the uncoated gold nanoparticle. The slope is not vertical because the nanoparticle is nonspherical but rather has large facets.⁹ The slope of the radial density plot for the coated gold nanoparticle is less steep than the uncoated nanoparticle. The slope decreases monotonically as more SAM chains are adsorbed onto the gold surface. This decrease in slope indicates that the atoms near the surface are more diffuse, resulting in the lower measured density of gold. The depth to which the gold core is affected we labeled as the surface corrugation.

In order to quantify the surface corrugation we defined by eq 7 a corrugation factor, C .

$$C = \frac{r(\rho = 0.85\rho_{\text{bulk}})_{\text{bare}} - r(\rho = 0.85\rho_{\text{bulk}})_{\text{coated}}}{r(\rho = 0.85\rho_{\text{bulk}})_{\text{bare}}} \quad (7)$$

In eq 7 $r(\rho = 0.85\rho_{\text{bulk}})$ is the radial position where the computed gold density is 85% of the density of bulk gold and the subscripts bare and coated refer to the uncoated and alkanethiolate SAM-coated gold nanoparticle, respectively. This definition of the corrugation factor will produce a value of 0.0 for the bare gold nanoparticle and maximum of 1.0 for a gold nanoparticle with impurities or a dissolved solute. For coated gold nanoparticles with various percentages of surface coverage the corrugation factor have been computed and are listed in Table 1.

The results presented in Table 1 show that the corrugation factor remains small, indicating little change in the nanoparticle

Table 1. Computed Corrugation Factors for Alkanethiolate SAM-Coated Gold Nanoparticles

coverage	C_{184} kJ/mol	C_{126} kJ/mol	$C_{\text{weak chain}}^a$
bare	0.000	0.000	N/A
10%	0.004	0.003	N/A
70%	0.012	0.018	N/A
80%	0.018	0.013	N/A
90%	0.043	0.020	N/A
100%	0.107	0.081	0.019

^a The gold–sulfur binding energy is 184 kJ/mol, but all chain–chain interactions are at 10%.

surface, for SAM coverage below 90%. Once the coverage reaches 90%, the corrugation factor increases 2.5 times over the value at 80% and increases by another factor of 2.5 times between 90% and 100%. With a corrugation value of 0.107, in column C_{184} kJ/mol, at full surface coverage the core of the gold nanoparticle is only about 89% of its original size. This result has two implications: first, the surface area of the gold nanoparticle has increased because the surface is no longer smooth but wrinkled; second, the SAM chains are able to form more bonds to the increased number of exposed gold atoms, potentially increasing the binding energy of the SAM to the gold nanoparticle. The results in Table 1 indicate an exponential dependence of the corrugation factor on the amount of surface coverage.

In the column for the lower sulfur–gold binding energy, namely, C_{126} kJ/mol, the corrugation is less pronounced, indicating a lower surface effect from the adsorbed SAM. The difference in magnitude of corrugation between the high and low binding energies is relatively small, about 25%. This small change in corrugation from a 33% change in gold–sulfur binding energy may indicate that the gold–sulfur binding energy is not the most important factor in determining surface corrugation. In the final column of Table 1 the gold–sulfur binding energy is 184 kJ/mol but all chain–chain interactions have been lowered to 10% of the previous magnitudes. Only the 100% surface coverage data is computed, but by comparing this result with the other columns it is apparent that the chain–chain interaction has an appreciable affect on the surface corrugation. The 90% drop in chain–chain interactions is accompanied by an 83% drop in corrugation. This is proportional to the change in corrugation observed by lowering the sulfur–gold binding energy alone.

There is experimental support for corrugation of the gold surface from the adsorption of an alkanethiol SAM.⁶⁰ In the experimental measurements it is assumed that an interior gold atom is pulled onto the surface to become an adatom. In this case the alkanethiol then bonds atop this gold adatom, and the gold surface is thus corrugated. Another consequence of this adatom is that the alkanethiol is more mobile and could be construed as a lower total sulfur to gold surface binding energy than if the sulfur atom were in the vicinity of many gold atoms. A rough estimate of the measured corrugation in this case would be the gold–gold bond distance divided by the nanoparticle radius, resulting in $C_{\text{exp}} = 0.116$, a result similar to the values in Table 1 for a 100% coated surface.

Phase Behavior and SAM Solubility

The computed potential energy (PE) of the alkanethiolate SAM-coated gold nanoparticle system is used to determine the critical temperature at which mixing and phase change will occur. The computed slope of the PE versus temperature curve from the MD simulation results are used to determine the heat capacity of the system.

In Figure 9 the system PE is plotted versus temperature for a 50 Å gold nanoparticle coated with increasing densities of the alkanethiolate SAM. For the bare nanoparticle there is a discontinuity around 980 K, which corresponds to a phase change. The SAM-coated nanoparticle demonstrates a very different behavior. Initially, for small numbers of adsorbed alkanethiolate chains, the melting temperature of the gold nanoparticle decreases and does not occur at a unique temperature but rather over a range of temperatures. This behavior is expected for a mixture of two materials.⁵⁶ The appearance of the SAM on the particle

(60) Yu, M.; Bovet, N.; Satterley, C. J.; Bengió, S.; Lovelock, K. R. J.; Milligan, P. K.; Jones, R. G.; Woodruff, D. P.; Dhanak, V. *Phys. Rev. Lett.* **2006**, *97*, 166102.

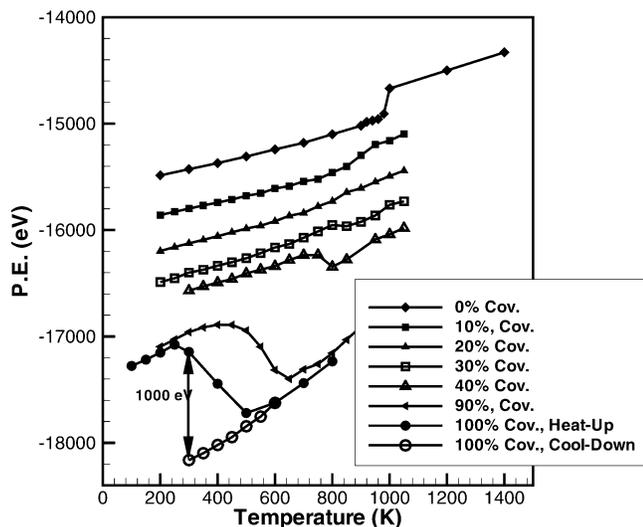


Figure 9. Potential energy versus temperature for the alkanethiolate SAM-coated gold nanoparticle with various amounts of surface coverage and the 184 kJ/mol sulfur binding energy.

surface decreases the surface tension and internal pressure. As the coverage percentage is increased further, another behavior is observed. At higher surface coverages, above 70%, the PE of the system begins to decrease as the temperature is increased and then after a small temperature range the PE resumes its linear increase. This behavior is explained by the mixing that occurs between the alkanethiolate chains and the gold atoms. The mixing of the SAM chains with the gold atoms is possible because of the lower binding energy of surface Au atoms. The lower binding energy of the surface atoms allows these atoms to become mobile at a lower temperature than the interior gold atoms.⁴⁹ The surface atoms begin to mix with the alkanethiol chains, and as the chains penetrate deeper into the nanoparticle they are able to interact with an everincreasing number of gold atoms. Once complete mixing has occurred, the PE resumes increasing monotonically.

As we found with the internal pressure in Figure 5, the structure of the SAM-coated gold nanoparticle is metastable. The increase of both the surface coverage and the temperature aids this particle in transforming into a mixed structure, Figure 10. This result has not been previously predicted as it was previously assumed that the SAM chains would desorb at these temperatures, as occurs on flat surfaces.¹⁷ If the SAM was to desorb, it is expected that the structure of the underlying gold nanoparticle would not be affected. This is reflected in the pair correlation data discussed earlier that shows that the gold nanoparticle with SAM coating is crystalline at a higher temperature with the lower 126 kJ/mol sulfur binding energy than with the 184 kJ/mol binding energy.

In Figure 11a the PE for the 126 kJ/mol sulfur binding energy simulation is plotted. During these simulations, as opposed to the 184 kJ/mol binding energy simulations, we observed some desorption of alkanethiol chains from the gold surface. This desorption process is irreversible because of the low density of desorbed chains in the simulation volume, a realistic assumption for coated nanoparticles in a vacuum or near vacuum. In Figure 11b it is interesting to note that the temperature at which the decrease in PE occurs for the 126 kJ/mol sulfur binding energy is higher than the 184 kJ/mol binding energy simulations. This result confirms the observation that when considered with a lower sulfur binding energy that the alkanethiol chains have a lower propensity to dissolve into the gold nanoparticle and a higher probability of desorbing from the gold surface.

The constant volume heat capacity, C_V , of the nanoparticle system is computed directly from the slope of the PE versus

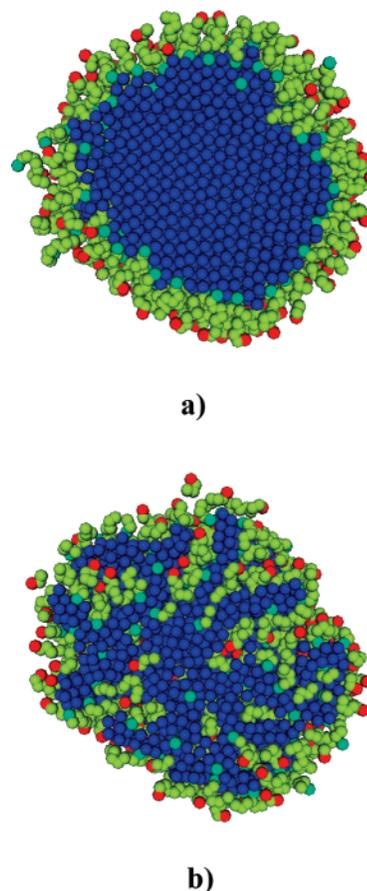


Figure 10. Cross section of 100% coated gold nanoparticle at 300 (a) and 600 K (b), showing the dissolution of the alkanethiol chains at high temperatures. In these images the blue spheres represent gold atoms, light blue represent sulfur, yellow represent CH_2 , and red represent CH_3 .

temperature curve. The portion of the PE versus temperature curve used to determine the heat capacity is that below the melting temperature or prior to mixing. For the nanoparticles studied in Figure 9, the specific heat capacity, c_V , is computed and plotted in Figure 12 for increasing amounts of surface coverage. In Figure 12 the computed heat capacity of the system per unit mass is increasing with increasing surface coverage. This result is reasonable since gold has a relatively low specific heat compared with the alkanethiolate chains. As more chains are added to the nanoparticle surface the fraction of the total system mass that they represent increases, resulting in the increase in the system heat capacity.

In addition to the heat capacity, the enthalpy of solution, H_{sol} , can be computed from the PE data. The enthalpy of solution is computed as the difference in system internal energy between the undissolved SAM chain system prior to heating and the internal energy of the same system after cooling down to the same temperature. The PE of the system decreases as a result of the mixing that occurs in the fully coated gold nanoparticle model during an increase in temperature from 300 to 600 K. This mixing may potentially occur completely at 300 K if it were possible to simulate long times (greater than 10 ns) in the MD simulations performed here. On the other hand, it is also possible that the activation energy for this transition to occur may be too high to complete the transition at 300 K. The measured change in system PE is 1000 eV, which is the enthalpy of solution.⁵⁶ Denoting this result per unit mass of the fully coated gold nanoparticle results in an enthalpy of solution of 100 kJ/kg. These results show that

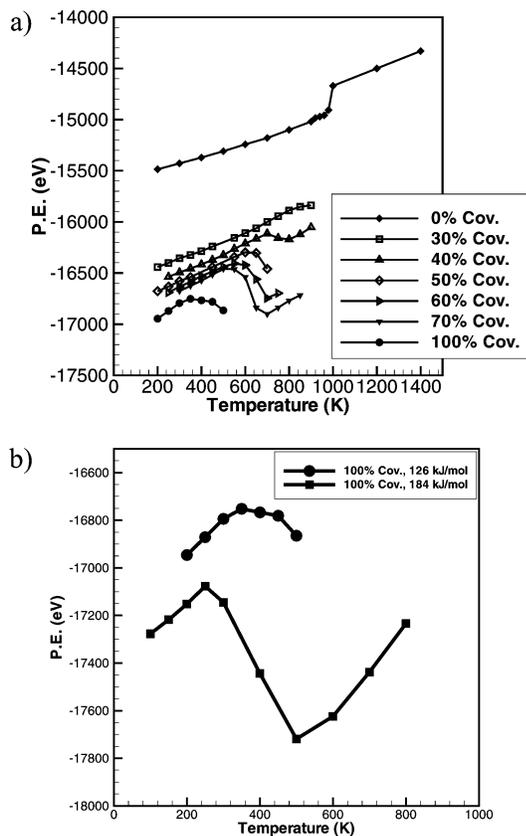


Figure 11. (a) Potential energy versus temperature for the alkanethiolate SAM-coated gold nanoparticle with various amounts of surface coverage and a 126 kJ/mol binding energy. (b) Comparison of potential energy versus temperature for a fully coated nanoparticle with 184 and 126 kJ/mol binding energies.

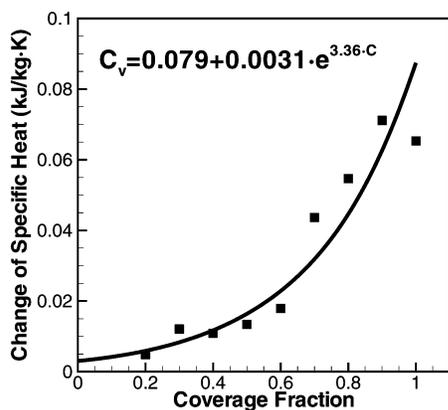


Figure 12. Plot of specific heat versus surface coverage for alkanethiolate SAM-coated gold nanoparticle showing the exponential relationship between specific heat and surface coverage.

the SAM has a measurable affect on not only the nanoparticle surface but also the entire system.

SAM Phases

Determining the phase diagram of the SAM chains on a gold surface is of fundamental interest. There has been extensive effort to understand the various phases of the alkanethiolate SAM chain on flat gold surfaces⁶¹ that have found four distinct phases. The four phases are as follows: a striped phase at low surface coverage, where the chains lie flat along the gold surface, intermediate structures for higher surface coverage, where some chains are

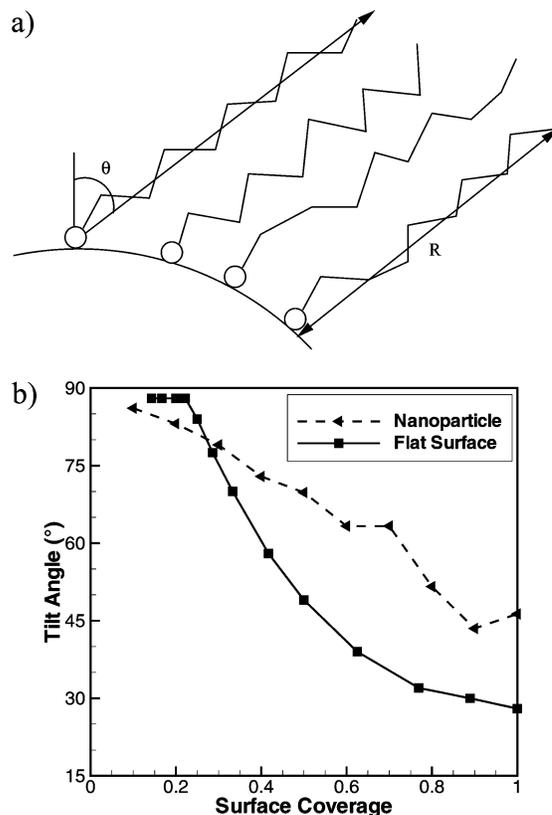


Figure 13. Difference in radial position for the head and tail groups of the alkanethiolate chains at various surface coverage densities: drawing (a) and tilt angle (θ) (b). The alkanethiol radius is also labeled as the monolayer thickness.

partially standing and others are lying flat. At high surface coverage the alkanethiolate chains are either in a highly structured $c(4 \times 2)$ phase⁷ at low temperatures or a more random liquid phase at higher temperatures. In this work, as in previous work involving flat surfaces, we investigated surface coverage and temperature as determining factors for the phase of the SAM.

It is expected that the binding energy of an alkanethiolate chain to the gold surface is dependent on whether the chain is lying along the surface of the nanoparticle (physisorption for longer chains or chemisorption for shorter chains) or standing radially outward with only the head group sulfur atom interacting with the gold surface (chemisorption).⁴⁵ The orientation of the chain is determined by measuring the tilt angle of the alkanethiol chain which is computed by comparing the radial position of the sulfur head group to the CH_3 tail group. For an alkanethiol chain lying along the nanoparticle surface the difference in radial position between the head and tail groups will be very small. When a chain is tilted upward the radius will be at a maximum value. This radius data is then used to estimate the angle that the chains make with the nanoparticle surface and measure the monolayer thickness.⁶²

In Figure 13b the tilt angle of the alkanethiol SAM is plotted for 10–100% surface coverage. At low surface coverage the alkanethiol chains lie along the surface of the nanoparticle as indicated by the almost 90° tilt angle. Between the high and low surface coverage regions the alkanethiolate chains begin to stand radially outward from the nanoparticle surface. Above 90% surface coverage the chains have all orientated themselves with the minimally observed tilt angle from the nanoparticle surface, and the maximum difference in radial position is observed. The

(61) Schreiber, F. J. *Phys.: Condens. Matter* **2004**, *16*, R881–R900.

(62) Hautman, J.; Klein, M. L. *J. Chem. Phys.* **1990**, *93* (10), 7483–7492.

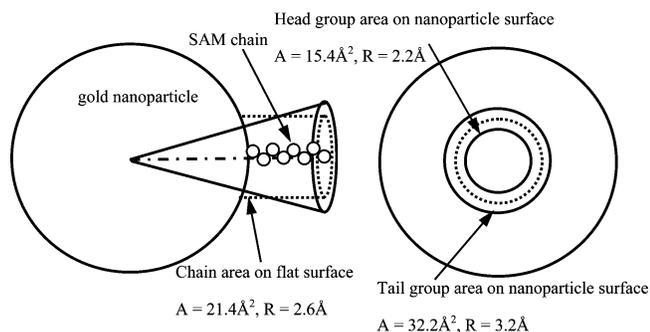


Figure 14. Drawing showing the difference in space occupied by the tail molecule on curved and flat gold surfaces, assuming a chain length of 9 carbon atoms.

tilt angle for the alkanethiolate SAM on a flat gold surface is also plotted for comparison. The steeper drop in angle and minimum tilt angle are both attributable to the geometric issues discussed next and illustrated in Figure 14.

In Figure 14 a side and front view of an idealized spherical nanoparticle is illustrated with an alkanethiol chain extending radially from the nanoparticle surface. The conic section in Figure 14 shows the volume associated with an alkanethiol chain adsorbed to the surface of a nanoparticle. The cylinder is the volume associated with an alkanethiol chain on a flat surface. Notice that the area of the cone sliced at the nanoparticle surface is smaller (15.4 \AA^2) than the area of the cylinder (21.4 \AA^2). This gives rise to the higher packing density on the nanoparticle surface compared to the flat gold surface. At the tail of the alkanethiol chain, for chains with 9 carbon atoms as considered here, the cone has a larger cross-sectional area than the cylinder. This larger area means that the tail of an alkanethiol chain has less restricted movement than the tail of an alkanethiol chain adsorbed onto a flat surface. This greater freedom of movement results in the higher tilt angle measured in this work (42°) compared to the tilt angle measured for a flat gold surface (about 30°).^{7,17}

Conclusions

The alkanethiolate SAM-coated gold nanoparticle has been shown to be mechanically and chemically metastable. The SAM that is adsorbed onto the surface begins to mix with the surface gold atoms at low temperatures because of the weaker gold–

gold bonding of the surface atoms to the interior gold atoms. This process is more pronounced as the density of the SAM is increased and as the gold–sulfur binding energy is increased. This is demonstrated by the corrugation numbers computed previously. For full surface coverage the SAM chains will begin to diffuse into the gold core as the temperature rises from 300 to 500 K. This mixing process lowers the PE of the system. The diffusion activation energy of the sulfur atoms is measured to be low in this temperature range. Once the temperature has increased to around 600 K, the alkanethiolate chains are completely mixed with the gold atoms and the activation energy for diffusion greatly increases. The PE and diffusivity results concur with this observation that a phase change in the SAM-coated gold nanoparticle system occurs around 500 K. We therefore concluded that the melting temperature of the gold is lowered by the adsorption of alkanethiolate SAMs on the nanoparticle surface and creates a metastable system for fully covered nanoparticles of diameters around 50 Å. The measured decrease in melting temperature of the nanoparticle is more pronounced as more SAM chains are adsorbed onto the nanoparticle surface.

We also compared each of the computed results for the low, 126 kJ/mol, gold–sulfur binding energy reported in the literature to the results from the high, 184 kJ/mol, binding energy. These comparisons show that the gold–sulfur binding energy does not have a large affect on the metastability of the nanoparticle. From these results we are able to conclude that the primary factor in determining if a nanoparticle will become metastable is the strength of the chain–chain interactions. Since experiments concur that desorption occurs in the 300–550 K temperature range we concluded that the 126 kJ/mol binding energy is most likely a better estimate of the true binding energy than the 184 kJ/mol value. Independent of the binding energy is the tilt angle, which is found to be greater for the nanoparticle geometry than for a flat surface.

Acknowledgment. The authors would like to acknowledge the support received by the U.S. Army Major Shared Resource Center (MSRC) at the Aberdeen Proving Ground, MD. Additional support was given by the National Institute of Standards Technology (NIST).

LA7024473